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Conductivity and Photoconductivity of a *p*-Type Organic Semiconductor under Ultra-Strong Coupling

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ABSTRACT: During the last decade, it has been shown that light-matter strong coupling of materials can lead to modified and often improved properties which has stimulated considerable interest. While charge transport can be enhanced in *n*-type organic semiconductors by coupling the electronic transition and thereby splitting the conduction band into polaritonic states, it is not clear whether the same process can also influence carrier transport in the valence band of *p*-type semiconductors. Here we demonstrate that it is indeed possible to enhance both the conductivity and photoconductivity of a *p*-type semiconductor rr-P3HT that is ultra-strongly coupled to plasmonic modes. It is due to the hybrid light-matter character of the virtual polaritonic excitations affecting the linear-response of the material. Furthermore, in addition to being enhanced, the photoconductivity of rr-P3HT shows modified spectral response due to the formation of the hybrid polaritonic states. This illustrates the potential of engineering the vacuum electromagnetic environment to improve the opto-electronic properties of organic materials.

KEYWORDS: strong coupling, organic semiconductors, conductivity, photoconductivity, *p*-type

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3 Light-matter strong coupling is a promising approach for controlling properties of matter.¹⁻²⁹ For
4 instance, it has been shown that charge transport can be enhanced by an order of magnitude in the
5 *n*-type perylene di-imide family of compounds and that the rate of energy transfer can be boosted,
6 leading to nearly unit transfer efficiency.¹¹⁻¹⁸ Furthermore, energy transfer can even be achieved
7 over distances well beyond what is expected from Förster theory by entangling donor and acceptor
8 to the same cavity mode.¹⁶⁻¹⁸ The modification of magneto-transport properties, with a reduction
9 in the longitudinal resistance, has recently been predicted³⁰ and reported for an ultra-strongly
10 coupled 2D electron gas.³¹

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12 The improved transport properties of organic materials in the strong coupling regime are the
13 result of the delocalized character of the collective states generated from the excited states or the
14 conduction band in the case of *n*-type semiconductors. However, the role of hybrid light-matter
15 states on the transport characteristics of *p*-type semiconductors, where valence band holes are the
16 majority carriers have not yet been studied, although it is also potentially important for organic
17 electronics.³²⁻³⁸ Interestingly, based on the theory of intersubband cavity polaritons,³⁸ one can
18 predict that when excitons are coupled to a confined optical mode and one reaches the ultra-strong
19 coupling regime, the ground state shifts while acquiring polaritonic character (*cf.* Supporting
20 Information (SI)).

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22 The fundamental question we explore here is whether this polaritonic character of the ground
23 state and its excitations leads to enhanced conductivity in *p*-type semiconductors. For that purpose,
24 we have studied a well-known *p*-type semiconductor, the regio-regular poly-(3-hexylthiophene)
25 (rr-P3HT) whose structure is shown in Figure 1a. Since photoconductivity plays a fundamental
26 role in many technological applications such as photodetectors, electrostatic imaging and
27 photovoltaics,³⁹⁻⁴⁵ we also explore whether the photoconductivity can also be improved in the
28 ultrastrong coupling regime.

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30 Strong light-matter coupling is achieved by placing a material with a well-defined transition in
31 a resonant electromagnetic field such as a surface plasmon or a Fabry-Perot cavity. Under the right
32 conditions, this strong interaction leads to the formation of two new hybrid light-matter states,
33 known as polaritonic states *P*⁺ and *P*⁻, as illustrated in Figure 1b, together with the so-called darks
34 states (DS). These states form under collective coupling when a single optical mode couples to a
35 large number of molecules as in these experiments. The energy separation between *P*⁺ and *P*⁻ is
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known as the Rabi splitting. In the ultra-strong coupling regime, which is the case here, the ground state is shifted downwards (Figure 1b) and it acquires polaritonic content as discussed above.

RESULTS AND DISCUSSION

The conductivity and the photoconductivity of the rr-P3HT were studied on Ag nano-hole arrays (having hexagonal periodicity) between drain and source electrodes, using the same

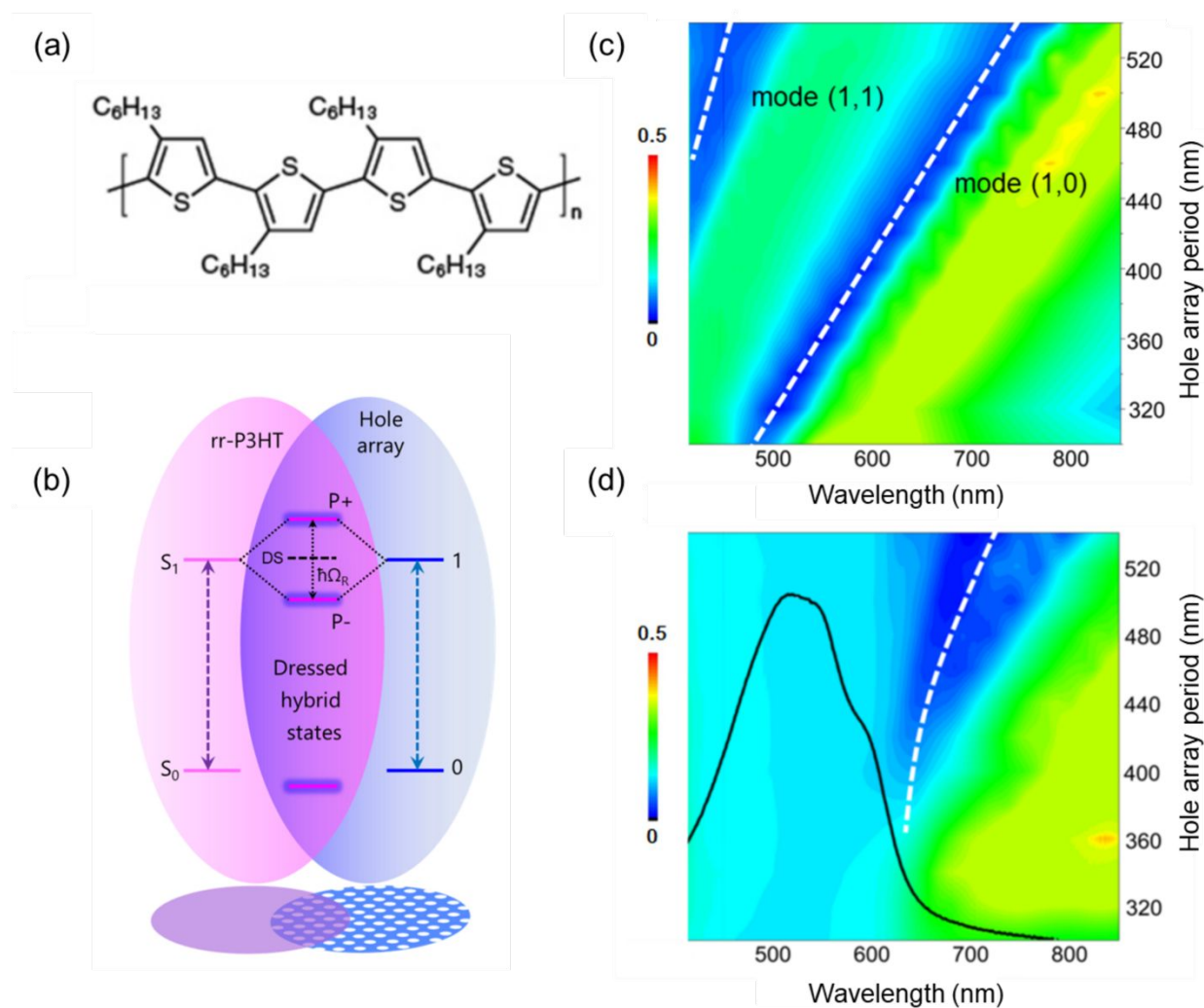


Figure 1. Strong coupling between surface plasmon and organic semiconductor: (a) Molecular structure of the *p*-type semiconductor rr-P3HT polymer. (b), Schematic representation of ultra-strong coupling between Ag hole array and rr-P3HT film. Contour image of transmission spectra of the hole arrays in the (c) absence and (d) presence of rr-P3HT (black curve shows the molecular absorption) as a function of period ranging from 320 nm to 540 nm (dashed white lines are guidelines to eyes) (x-axis data is not shown below 415 nm, the lower wavelength limit of our optical spectrophotometer detection).

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3 approach as in ref. 11. Ag was chosen for its capacity to sustain low-loss surface plasmon modes
4 in the wavelength range of interest while still providing good injection capacity for the electrical
5 measurements.^{11,33} As detailed in the Experimental section, the hexagonal hole arrays were milled
6 with a Zeiss Auriga dual beam FIB/SEM in a 100 nm thick Ag film and the period P of the holes
7 were varied (from 320 nm to 560 nm) in such a way that the plasmonic resonances (Figure S1, SI)
8 are in the same range as the first absorption bands of the semiconductor. Surface plasmon
9 resonances (λ_{sp}) of hexagonal hole arrays having different periodicity (P) are given by:

$$\lambda_{sp} = \frac{P\sqrt{3}}{2} \sqrt{\frac{\epsilon_m \epsilon_d}{\epsilon_m + \epsilon_d}}$$

10 where, i and j are integers; ϵ_m and ϵ_d are the refractive index of metal and dielectric material
11 respectively. The different modes of the hole array is denoted by the indices i and j as (i,j). Thus
12 Ag hole arrays of given periodicity (320 nm to 560 nm) exhibit corresponding (1,0) and (1,1)
13 surface plasmon modes in the visible region, which can be tuned to the absorption band of the
14 semiconductors. The organic semiconductor rr-P3HT was prepared and purified as described in
15 the literature.⁴⁶ It was then dissolved in trichloroethylene by Soxhlet extraction method and spin-
16 coated on the hole arrays, yielding *ca.* 85 nm thick films. The sample was then covered with a 200
17 nm PVA (poly vinyl alcohol) film. The strong coupling regime was verified by measuring their
18 transmission spectra for different hole array periods P and comparing them with the bare hole
19 arrays (*i.e.* without absorber) (Figure 1c and d). We observe a mode bending of the lower
20 polaritonic branch (*P*-) in the presence of organic semiconductor. *P*+ is in the UV region and
21 cannot be detected with optical microscope with which the spectra were taken. The resulting Rabi
22 splitting are estimated to be *ca.* 0.6 eV for rr-P3HT corresponding to 27% of the transition energy
23 as explained in the Supporting Information and in agreement with the theoretical analysis
24 presented below where all the polaritonic branches can be seen. Such large Rabi-splitting indicates
25 that the system is in the ultra-strong coupling regime where all the states of the system are modified
26 with significant consequences as we will discuss further down.

27 The conductivity of the samples was then characterized using the setup schematically shown in
28 Figure 2a. The I-V (current-voltage) curves of the various samples were then recorded as shown
29 in Figure 2b as a function of the hole period over a voltage range (-5 to 5 V). From the measured
30 I-V curve, current enhancement factor was calculated by comparing the current in the periodic
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hole array systems with that of a random hole array. Figure 2c plots the current enhancement factor at ± 5 V as a function of $2\pi/P$ for dark current of rr-P3HT coated hole arrays

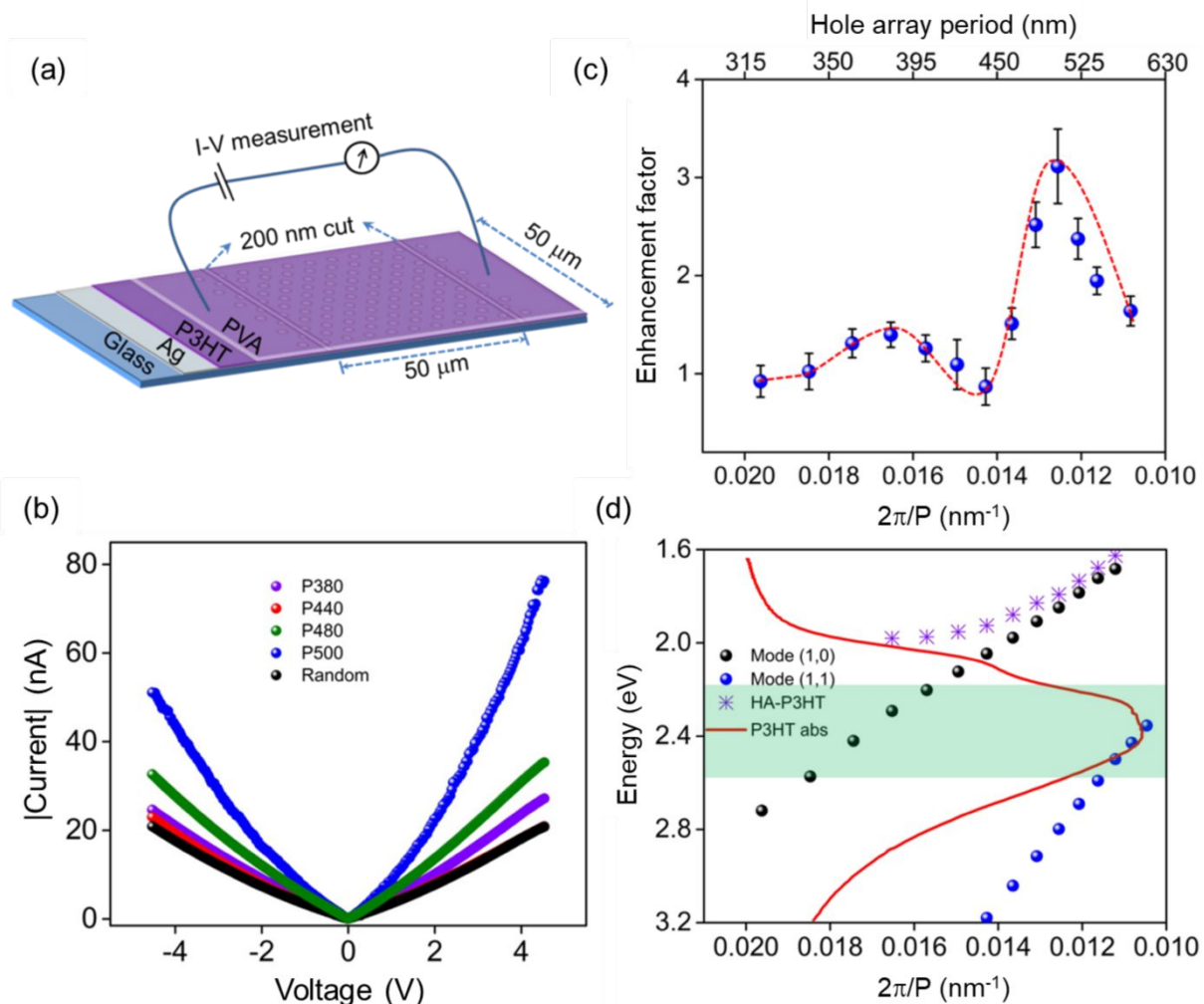


Figure 2. Current-Voltage measurement: (a) Schematic representation of the set-up used to perform conductivity measurement. (b) Current-voltage (I-V) curves of the rr-P3HT samples as a function of the period P of the hole arrays. (c) Conductivity enhancement with respect to the in-plane momentum of hole arrays (top axis shows corresponding hole array period in nm). Blue dots show the average value of three different experiments and the error bars show the standard deviation. Red dashed line is a guide to the eye. (d) The conductivity enhancement observed in c, correspond to the intersection of the rr-P3HT molecular absorbance (red line) and the surface plasmon resonance for the (1,0; black dots) and (1,1; blue dots) modes of hexagonal hole arrays in Ag; The violet stars show the lower polaritonic band. The broad absorption region of rr-P3HT is highlighted.

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3 with respect to a random array. The current enhancement factor peaks when the absorption
4 maximum of rr-P3HT crosses the (1,0) and (1,1) modes of the hexagonal plasmonic array (Figure
5 2d) and undergoes strong coupling. In other words, this occurs only when the plasmonic array is
6 in resonance with the rr-P3HT absorption. The small differences, 3% and 8%, between the
7 crossings of the (1,0) and (1,1) modes with the rr-P3HT absorption (Fig. 2d) and the current peaks
8 (Fig. 2c) is well within experimental error when considering the following. The surface plasmon
9 mode position in the bare Ag hole arrays can only be calculated in the absence of the absorber.
10 Obviously, the refractive index and therefore the surface plasmon modes will be modified by the
11 presence of the absorber even in the absence of strong coupling. In addition, the rr-P3HT
12 absorption is so broad that it overlaps with both modes simultaneously. When compared to the
13 conductivity on a random hole array, rr-P3HT on resonance hole array ($P=500$ nm; $k_x=0.0126$ nm⁻¹)
14 exhibits an enhancement factor of ~ 3 for the dark current measurement. This enhancement is
15 smaller than the maximum observed for a *n*-type perylene di-imide semiconductor where it reached
16 a factor of 10. As we demonstrated by comparing three different *n*-type semiconductors, the
17 disorder in the material appears to have a large negative effect on the enhancement.¹⁵ So one
18 possible explanation for lower enhancement in rr-P3HT, as compared to the best *n*-type PDI
19 semiconductor we have tested, is the higher disorder.
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32 A number of control experiments were carried out in order to exclude possible artifacts. The
33 presence of oxygen can modify the conductivity of rr-P3HT compounds so we checked that the
34 results under nitrogen atmosphere for our coated samples were the same within experimental error.
35 The metal substrate is floating and does not contribute to the conductivity due to the Schottky
36 barrier at the semiconductor-metal interface. This was confirmed by depositing a 20 nm insulating
37 SiO_x layer between the metal hole array and organic semiconductor which gave very similar results
38 (Figure S2, SI). It was also verified that the observed boosts in current were not due to the cuts
39 that are made in the hole arrays. As shown in the SEM image of the array after the cutting (Figure
40 S3a, SI), the edge is smoothed by the FIB and finally random hole arrays, which have no
41 pronounced plasmon mode, displays no enhancement (Figure S3b, SI).
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50 Enhanced conductivity in a *p*-type semiconductor under strong coupling indicates that the charge
51 carrying valence band is also affected by the presence of polaritonic states even in the absence of
52 illumination. As mentioned earlier, the system is in the ultra-strong coupling regime under
53 collective coupling conditions where counter-rotating and diamagnetic terms
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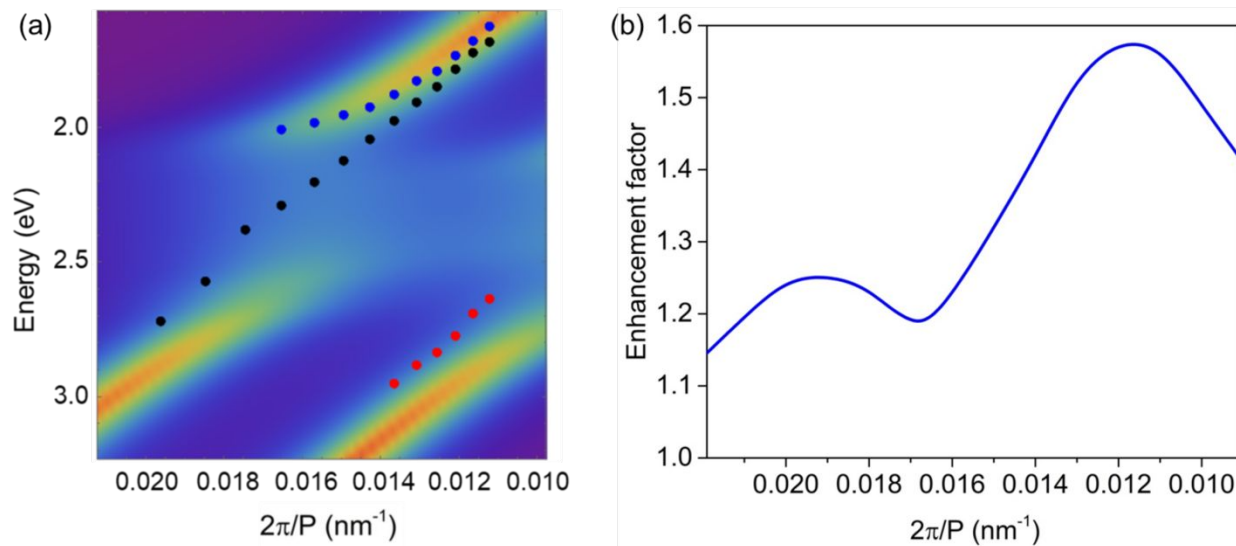


Figure 3. Results of the theoretical modelling: (a) the contour plot depicts the calculated photonic component of the polariton modes (cf. Supporting Information) as a function of frequency and $2\pi/P$. Black and red dots are experimental data for the bare (1,0) and (1,1) resonator modes respectively. The blue dots are the experimentally measured polariton dispersion. (b) Conductivity enhancement *versus* $2\pi/P$. In the calculation, the exciton transition energy is taken equal to 2.35 eV with an inhomogeneous broadening equal to 0.3 eV. The vacuum Rabi splitting is 0.6 eV. For the bare electronic scattering time, we have taken $\omega_{exc}\tau_0 = 0.1$ where $\hbar\omega_{exc}$ is the exciton transition energy.

cannot be neglected. The effects of the light-matter coupling on the conductivity can be predicted *via* a many-body Kubo linear-response approach.^{30, 47} In this approach, the conductivity is simply given by a current operator \hat{J} evaluated between the collective many-body ground state $|GS\rangle$ (all molecules in their respective ground states S_0) and the polaritonic states $|r\rangle$ as $|\langle GS|\hat{J}|r\rangle|^2$, where only the polariton modes that have a zero in-plane wavevector component contribute. This linear-response approach therefore yields directly a conductivity, and thus transport properties of the material, that crucially depend on the hybrid light-matter nature of the ground state and of the virtual polaritonic excitations affecting the transport *via* a renormalization of the current response and scattering times, as detailed in the Supporting Information. Such a linear response approach includes all virtual polariton excitations, but in this work, we have simplified the model and have calculated the conductivity mediated by virtual polariton excitations by considering only two cavity modes and one disordered molecular ensemble with a Gaussian spread of the excitonic transition frequency. The value of the vacuum Rabi coupling and of the inhomogeneous broadening of the excitonic mode can be determined by fitting the cavity mode frequency

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3 dispersions, the available polariton dispersion and the exciton absorption spectrum, as depicted in
4 Figure 3a. The only left adjustable parameter in the model is the Drude scattering time τ_0
5 (associated to the mobility of the organic semiconductor) that can be determined by comparison
6 to the experimental conductivity curves.⁴⁷
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10 The value of the scattering time that best fits the experimental behavior is consistent with the
11 typical relatively-low mobilities of the considered organic semiconductors. In the $2\pi/P$ range of
12 this study, this rather simple model predicts, as shown in Figure 3b, a conductivity enhancement,
13 with two asymmetric peaks whose relative intensities and positions are rather close to the
14 experimental values in Figure 2c. For the considered parameters, the theoretical peaks are broader
15 than in the experiments and the maximum enhancement is around 1.6 instead of the measured
16 value around 3. This difference can be ascribed to several factors, such as the actual disorder
17 configuration of the experimental sample, which would require a much more sophisticated
18 theoretical modeling, which is beyond the scope of this work.
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26 We now turn to the photoconductivity measurements. Interestingly, the photoconductivity of rr-
27 P3HT benefits from the enhanced conductivity (Figure S4, SI) under strong coupling in the same
28 way as the dark current does, although the absolute values are higher under illumination (Figure
29 4). The photoconductivity also peaks for the resonant hole array ($P=500$ nm; $k_x=0.0126$ nm⁻¹) with
30 an enhancement factor of ~ 3 , when compared to that of on a random hole array (Figure S4, SI).
31 Figure 4a shows the linear dependence of the photocurrent on the excitation intensity for rr-P3HT
32 when measured under white light cw (continuous wave) illumination. At such low intensities per
33 unit area (<15 mW per cm²), the fraction of excited state absorbers remains at all time negligible
34 and as a consequence there is no significant depopulation of the ground state and the collective
35 Rabi splitting is unmodified. The quantum yield for the photogeneration of charge carriers is
36 estimated to be ca. 5×10^{-3} for rr-P3HT (520 nm illumination). The spectral response of the
37 photocurrent was also studied using a tunable pulsed laser source (NKT laser) under strong
38 coupling conditions and compared to that of a random hole array film. The ratio of the current for
39 these two conditions is shown in Figure 4c and reveals enhanced photocurrent at longer
40 wavelengths than the absorbance peak of the bare rr-P3HT with a maximum around 620 nm. This
41 corresponds to the P^- peak. Since the spectral enhancement at long wavelengths is due to splitting
42 of the original peak into P^+ and P^- , a similar enhancement is expected to be seen at shorter
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wavelength corresponding to the P^+ transition but our tunable laser system cannot reach the corresponding wavelengths.

For purposes of comparison, we also studied the photoconductivity of a perylene di-imide (PDI) n -type semiconductor under strong coupling (Figure S5, SI). We have already shown that this class of compounds exhibits an enhanced conductivity under strong coupling.¹¹ The compound used here is a slightly different derivative whose structure is shown in the Supporting

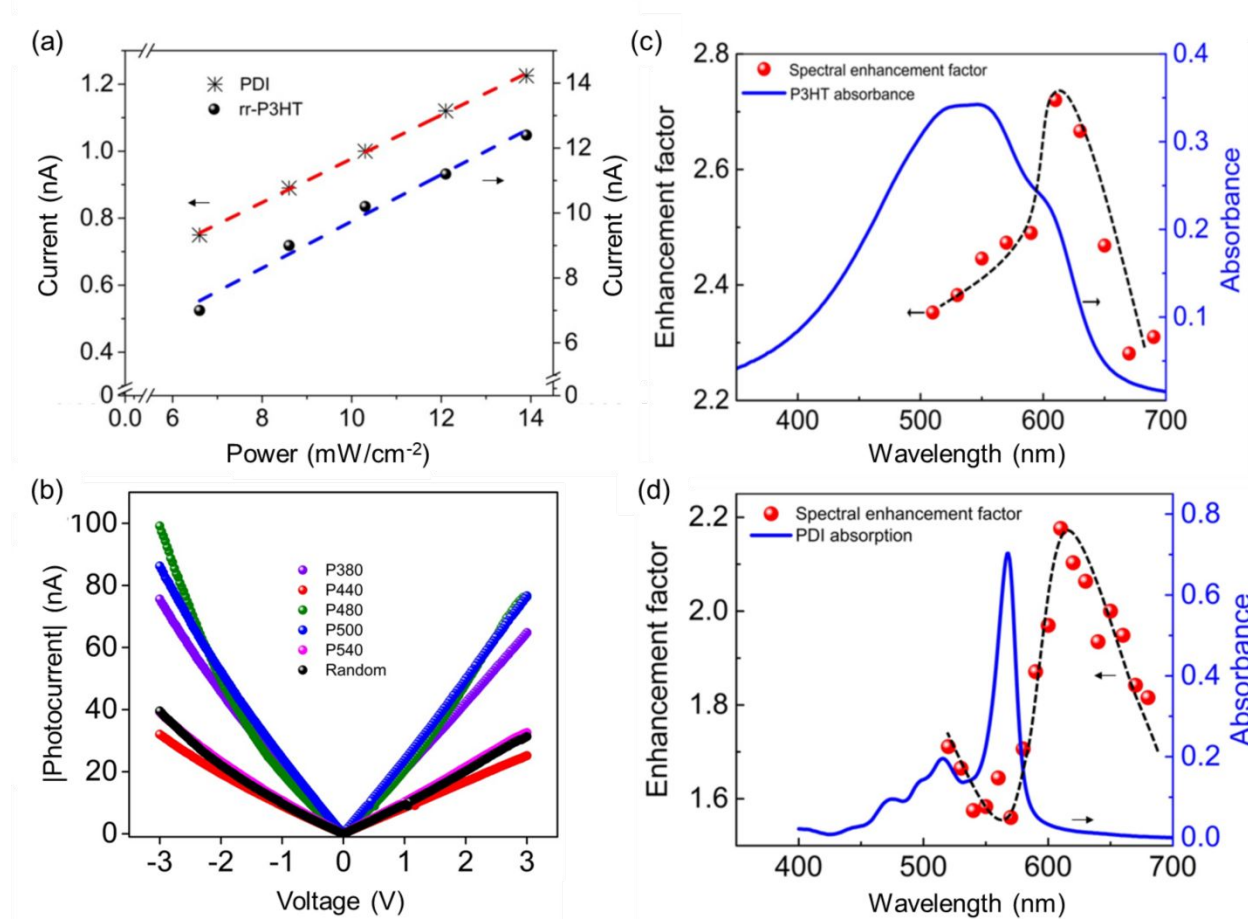


Figure 4. Photoconductivity under ultra-strong coupling: (a) Photocurrent as function of the white light illumination power for rr-P3HT and PDI under strong coupling ($P=500$ nm and $P=280$ nm respectively). (b) Current-Voltage (I-V) curves of rr-P3HT samples as a function of the period of the hexagonal hole array under white light illumination (47 mW cm⁻²). (c) Spectral response under ultra-strong coupling (red dots, $P=500$ nm) as a function of excitation wavelength, the enhancement factor is the ratio of the photocurrent divided by the corresponding photocurrent of an uncoupled (random array) rr-P3HT film. A tunable pulsed laser with a 10 nm bandpass was used for this study. The peak at 620 nm corresponds to the P^- peak. The blue curve gives the absorbance spectrum of the bare rr-P3HT film. The black lines are guides to the eye. (d) Idem for PDI under strong coupling ($P=280$ nm), the peak at 610 nm corresponds to P^- peak.

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3 Information. Figure S5 (SI) summarizes the results. We observe that the photoconductivity of PDI
4 is also enhanced upon ultra-strong coupling (Rabi splitting is 34% of transition energy, see
5 Supporting Information) with two current peaks appearing at the crossing of the PDI film
6 absorbance peak and the plasmonic modes. Figure 4a shows the linear dependence of the current
7 with illumination power and Figure 4d shows that just as for rr-P3HT, the spectral response of PDI
8 is enhanced where *P*- absorbs.
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15 CONCLUSIONS

17 In conclusion, using a *p*-type organic semiconductor, we have demonstrated that hole transport
18 in the ground state can also be enhanced by strong coupling. This is in turn a direct evidence of
19 the polaritonic character of the valence band, in agreement with theoretical predictions on material
20 properties under ultra-strong coupling.³⁸ We have also shown that the photoconductivity, like
21 conductivity, can be boosted by the formation of the delocalized polaritonic states and in addition
22 their spectral response can be extended and tailored by the coupling strength. This works for both
23 *n*- and *p*-type semiconductors. A recent study⁴⁸ has come to our attention where an enhanced
24 photoconductivity in a *p*-type organic semiconductor was demonstrated under strong coupling with
25 a Fabry-Perot cavity mode, confirming that charge transport in organic semiconductors can be
26 modified by the ultra-strong light matter hybridization in various configurations. Such findings
27 could stimulate more detailed analysis of photoconductivity by time resolved pump-probe
28 techniques which will provide insight into the dynamics of charge carriers in the strong and ultra-
29 strong coupling regimes. Strong coupling is easy to implement in a variety of configurations. We
30 therefore expect that such results will find applications in devices where photoconductivity plays
31 a role, in particular in photovoltaics and photodetectors. More broadly speaking, taken together
32 with other investigations, it shows the potential of engineering the electromagnetic vacuum
33 environment for material science and devices.
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48 EXPERIMENTAL SECTION

49 Synthesis of materials:

51 Regioregular poly(3-hexylthiophene), rr-P3HT was synthesized and purified as reported
52 earlier.⁴⁶
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55 Two-terminal device with nanohole arrays for photoconductivity measurements:

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3 All the 2-terminal electrodes were realized on a glass substrate BK7 (25×25 mm) after standard
4 cleaning procedures with Hellmanex™ III solution (1.0% solution in milliQ water) in a 35 kHz
5 sonication bath, then rinsed with water and sonicated for 1 h in (spectroscopically pure) ethanol.
6 The glass substrates were then dried in an oven covered with Aluminium foil. 100 nm thick Ag
7 electrodes were fabricated using a metallic cross mask containing 12 fingers (approx. 50 μm width)
8 in an electron beam evaporator (Plassys ME 300) at optimized working pressure ($\sim 10^{-6}$ mbar) and
9 deposition rates (~ 2 nm s⁻¹). Plasmonic hole arrays were generated by the NPVE software program
10 and milled using a Carl Zeiss Auriga FIB system. 1.0 wt% of rr-P3HT and PDI solutions were
11 freshly prepared by dissolving the molecules in spectroscopic grade trichloroethylene (Soxhlet
12 extraction) and anhydrous chloroform respectively at ambient conditions and spin-coating onto the
13 electrodes at 1000 r.p.m., to achieve 85 nm-thick films. The reddish, smooth rr-P3HT and PDI
14 thin-films were completely dried and annealed on a hot plate above the glass transition
15 temperature. A passive layer of PVA was spin-coated (~ 200 nm) further to protect the active layer
16 to avoid direct exposure to air and moisture. For a test experiment, in order to introduce an
17 insulating layer in between the metal hole array and organic semiconductor, 20 nm SiO_x was
18 sputtered on top of the metal hole array by using magnetron sputter under reduced pressure (10
19 mTorr) at 200 W for 50 sec. On top of the sputtered SiO_x, organic semiconductor and PVA layer
20 were spin coated as explained above.
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34 The electrical characterization of the 2-terminal device was carried out by means of a Cascade
35 Microtech MPS-150 probe station equipped with micro-positioners to contact the electrode pads.
36 Both the dark current and photoconductivity I–V characteristics are recorded by means of a
37 Keithley 2636B source meter interfaced with Labtracer 2.0 software. Photocurrent generation was
38 tested under white light cw illumination and with a super-continuum ps pulsed laser (NKT
39 Photonics) with a nominal power of 100 mW to 200 mW, with variable wavelength (500-800 nm;
40 10 nm FWHM) provided by a SuperK tunable single line filter. The output of the FC/PC fiber was
41 directed to the center of the electrode covering a spot size of 2 mm in diameter.
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50 ASSOCIATED CONTENT

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52 A previous version of this manuscript has been submitted to a preprint server.⁴⁹ Theoretical
53 analysis for the dark conductivity, synthesis of PDI, microscopic characterization of Ag hole
54 arrays, sample fabrication and I-V measurement with an insulating layer, SEM image of periodic
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and random hole arrays, photo-conductivity of PDI and rr-P3HT under strong coupling. The Supporting Information is available free of charge at <https://pubs.acs.org>.

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AUTHOR CONTRIBUTIONS

The manuscript was written through contributions of all authors. All authors have given approval to the final version of the manuscript.

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